

# **Polymers with “Goldilocks” Cavities Tuned for Fast, Selective Transport of Small Molecules and Ions**

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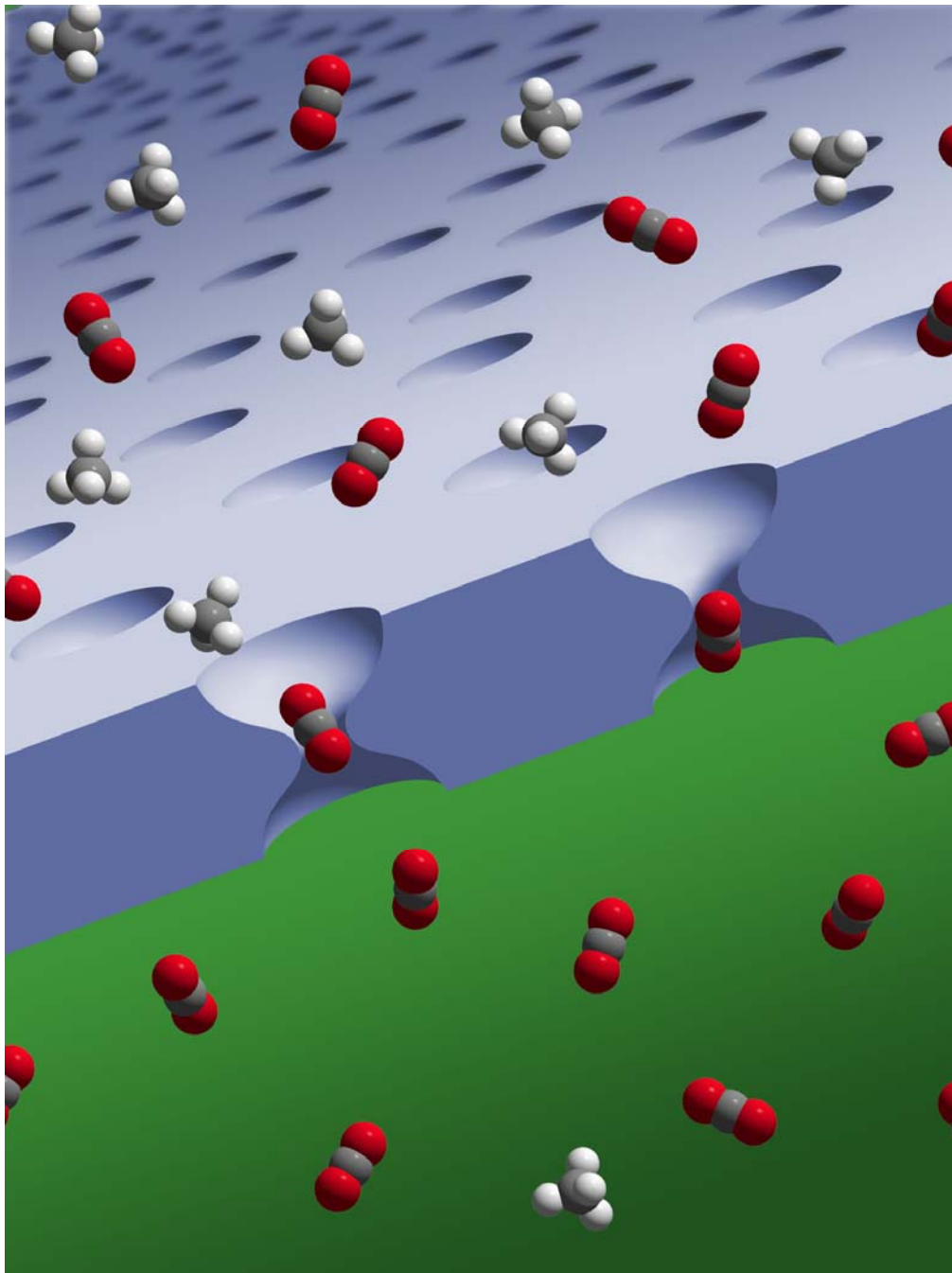
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"Hourglass" shaped Membrane

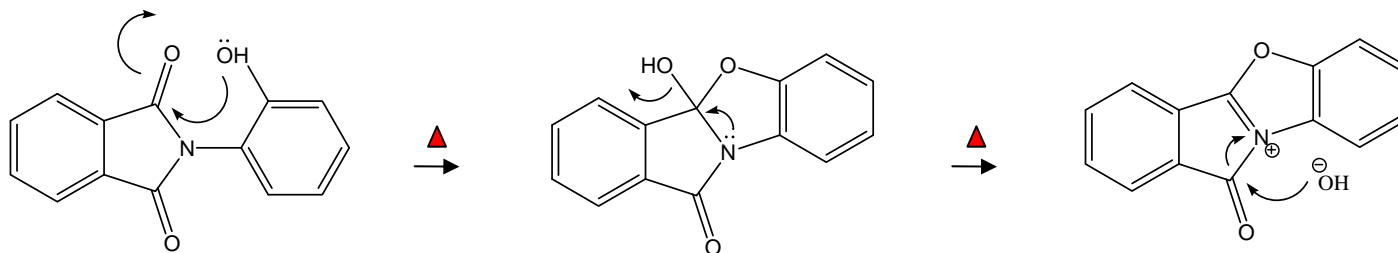
*Science* 318, 254 (2007)

# Candidate Polymers

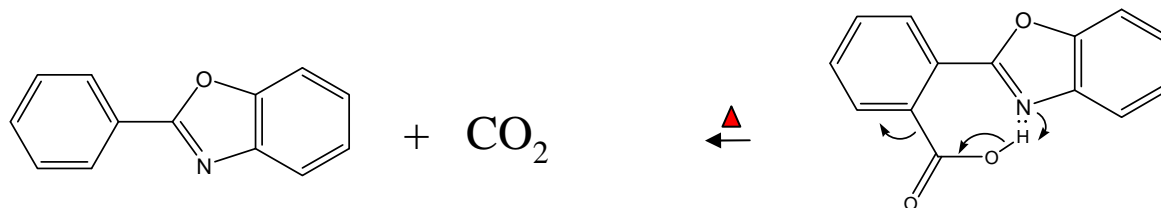
- *Polybenzimidazoles*
- *Polyquinoxalines*
- *Polybenzothiazoles*
- *Polybenzoxazoles*

- High temperature durability (~600 °C at air atmosphere)
- High chemical stability (soluble in strong acids)
- High mechanical property
- Flat, rigid rod structure for good gas selectivity
- Strong interchain interaction due to charge transfer complex
- **Low gas permeability and difficult membrane fabrication**
- **Not enough gas permeation data**

# Thermal Conversion Mechanism Of PIOFG



- Temperature: 350 – 500 °C
- Atmosphere: vacuum or inert
- State: Film, fiber, and powder (solid state)



1. *Vysokomol. Soyed.* B9 (1967) 873
2. *Polymer* 40 (1999) 3463

# Prerequisites for New Polymer Designs

## **For high permeability,**

- > fast diffusion and large solubility – large free volume elements
- > analog of microporous inorganic materials

## **For high selectivity,**

- > narrow cavity size distribution
- > analog of amorphous carbon molecular sieve

## **For excellent thermal and chemical stability,**

- > analog of semi-rigid or rigid-rod aromatic polymers

## **For slow physical aging**

- > strong interchain interaction, or crosslinking structure

# Polymer Membranes for Gas Separation

$$P = D \cdot S \quad (1)$$

*Solution-diffusion mechanism*

$$\alpha_{A/B} = \frac{P_A}{P_B} = \left( \frac{D_A}{D_B} \right) \cdot \left( \frac{S_A}{S_B} \right) \quad (2)$$

- *Diffusion-selective*

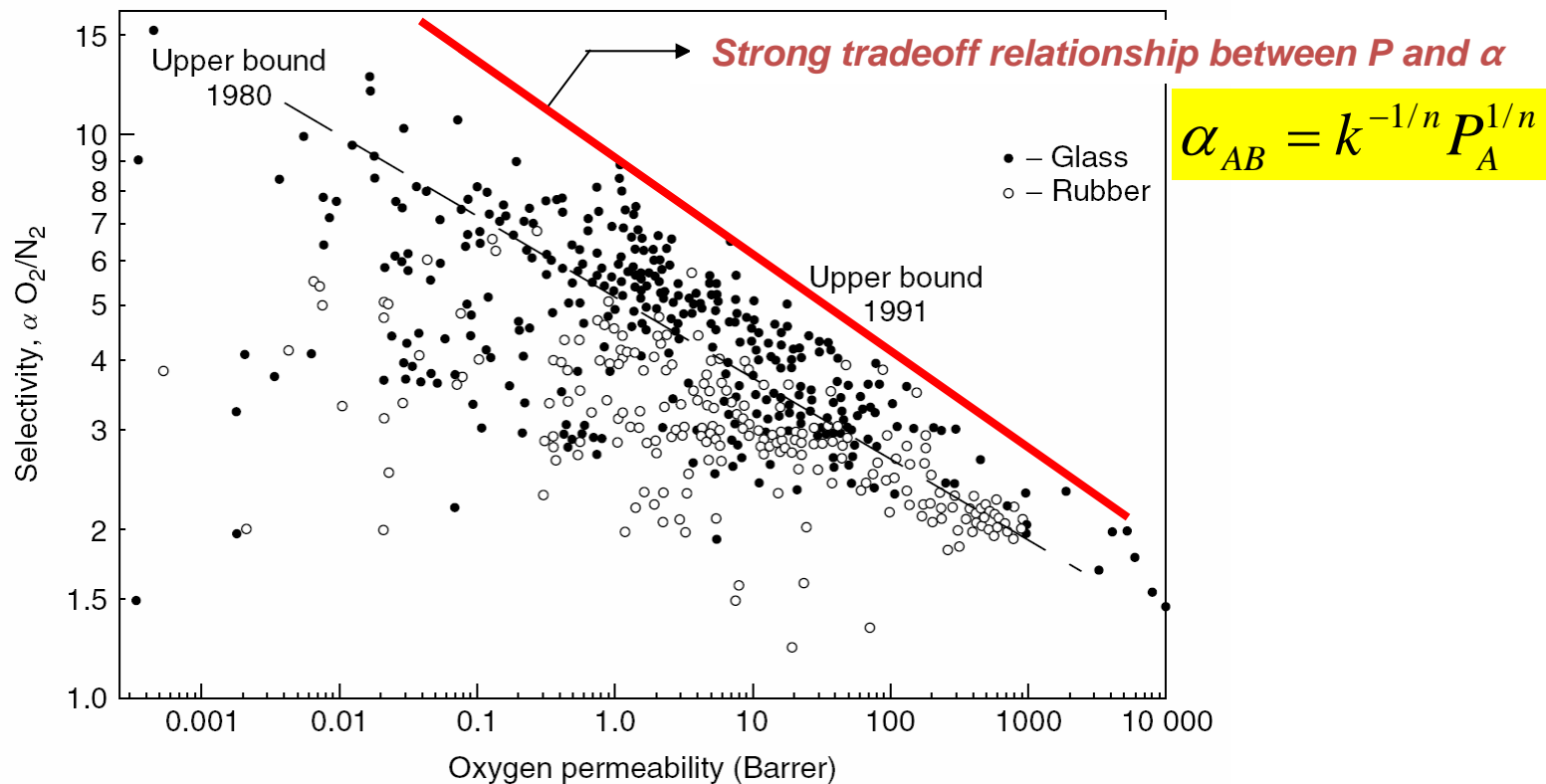
- O<sub>2</sub>/N<sub>2</sub>, H<sub>2</sub>/N<sub>2</sub>, H<sub>2</sub>/CH<sub>4</sub>, etc
- Glassy polymers (rigid & stiff chain)

- *Sorption-selective*

- C<sub>>3+</sub>/H<sub>2</sub>, CO<sub>2</sub>/H<sub>2</sub>, Vapor/permanent gas, etc
- Rubbery polymers

# Current Limitation of Polymeric Membranes for Gas Separation

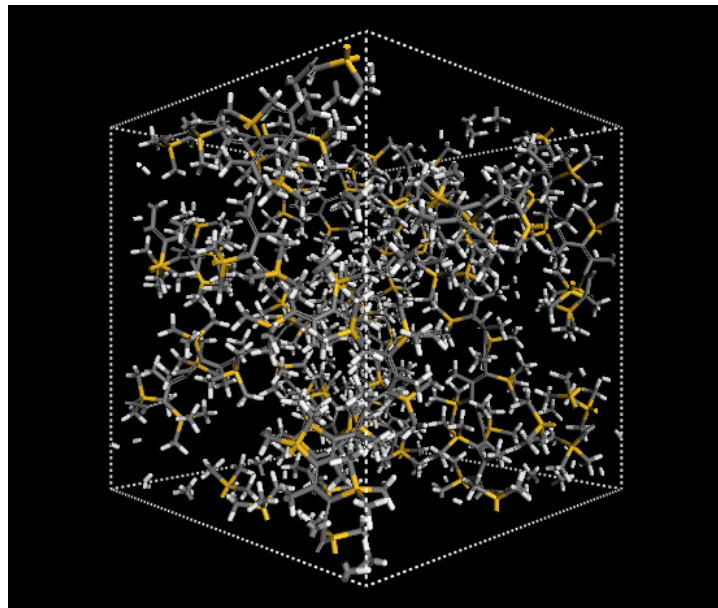
Robeson, L.M., J. Membr. Sci. 62, 165 (1991)



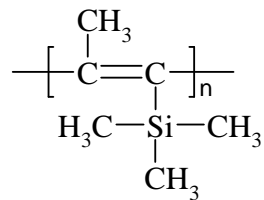
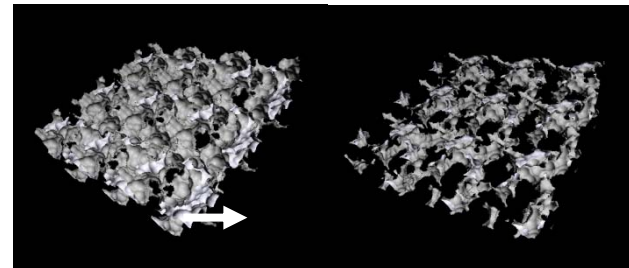
*Usually, glassy polymer membranes show high selectivity,  
but low permeability...*

# The Most Permeable Glassy Polymer (PTMSP)

PTMSP



- *High glass transition temperature*
- *High free volume*
- *Ultrahigh gas permeability*
- *Low selectivity for small gases*
- *Fast physical aging*



Bulk density  
= 0.75 gcm<sup>-3</sup>

→  
aging

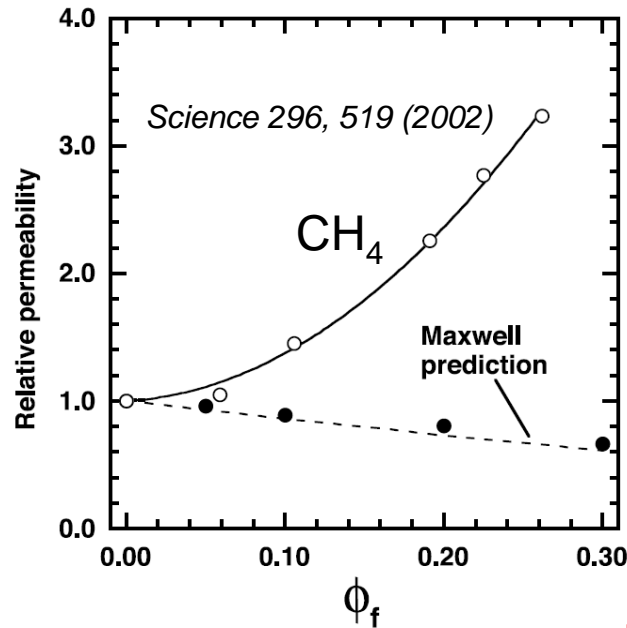
Bulk density  
= 0.95 gcm<sup>-3</sup>

O<sub>2</sub> permeability: 6000 Barrer  
O<sub>2</sub>/N<sub>2</sub>: 1.8

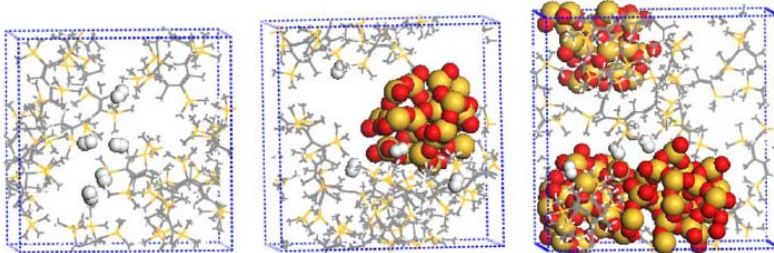
O<sub>2</sub> permeability: 1820 Barrer  
O<sub>2</sub>/N<sub>2</sub>: 1.9

# Way to Improve Gas Permeability in Glassy Polymer Membranes

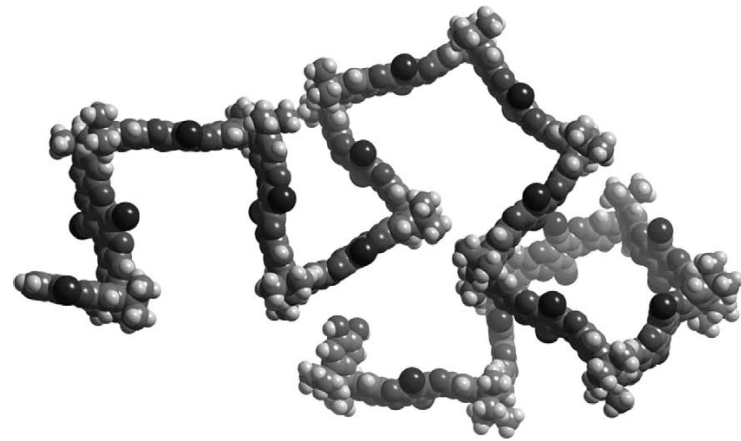
## (1) Nanocomposite



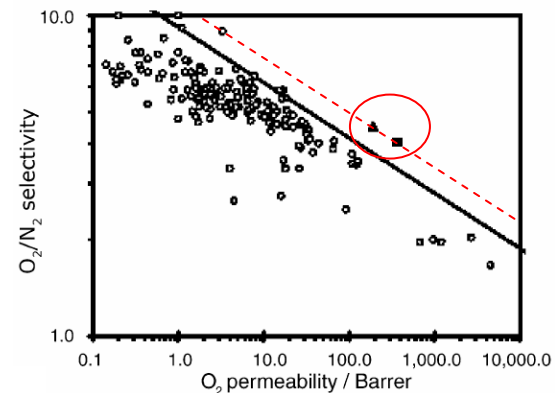
Volume fraction of fumed silica



## (2) Polymer with intrinsic microporosity

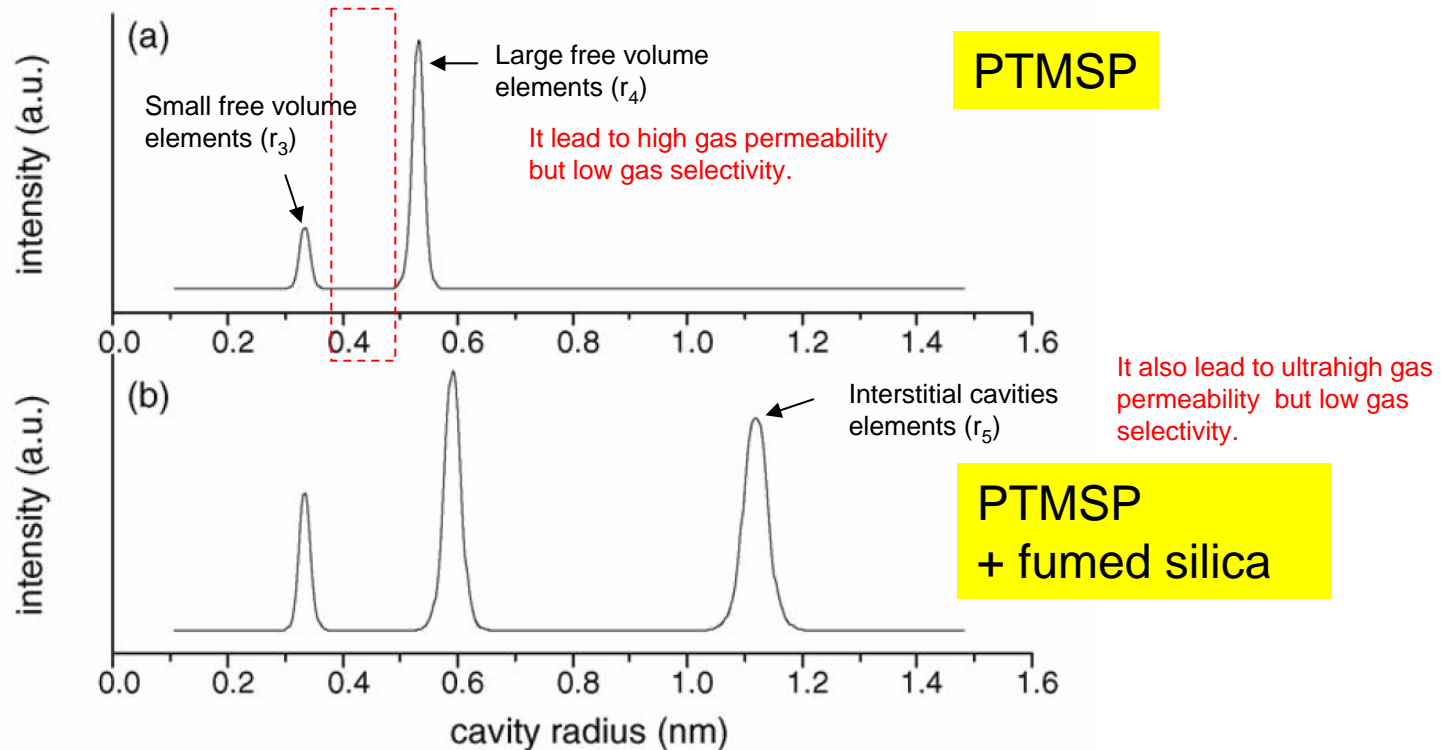


*J. Membr. Sci.* 251, 263 (2005)

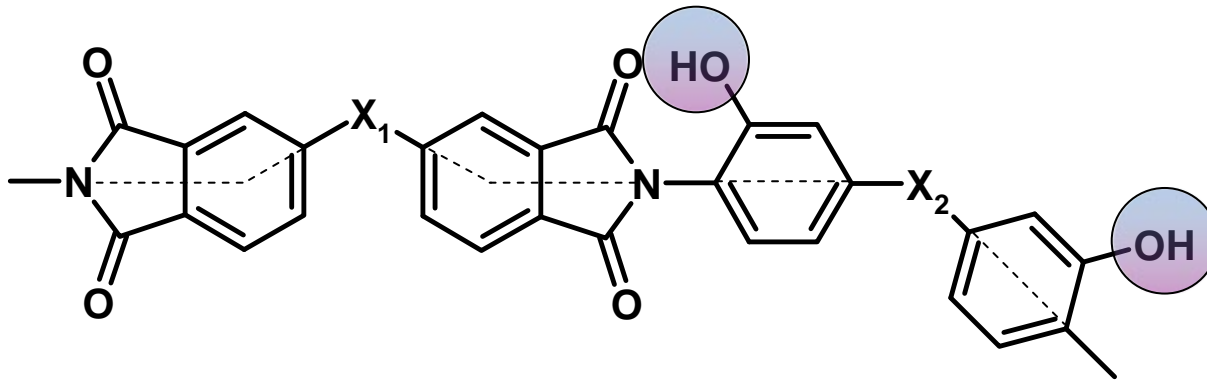


# Free Volume Elements for Fast Molecular Transport

## Region of interest

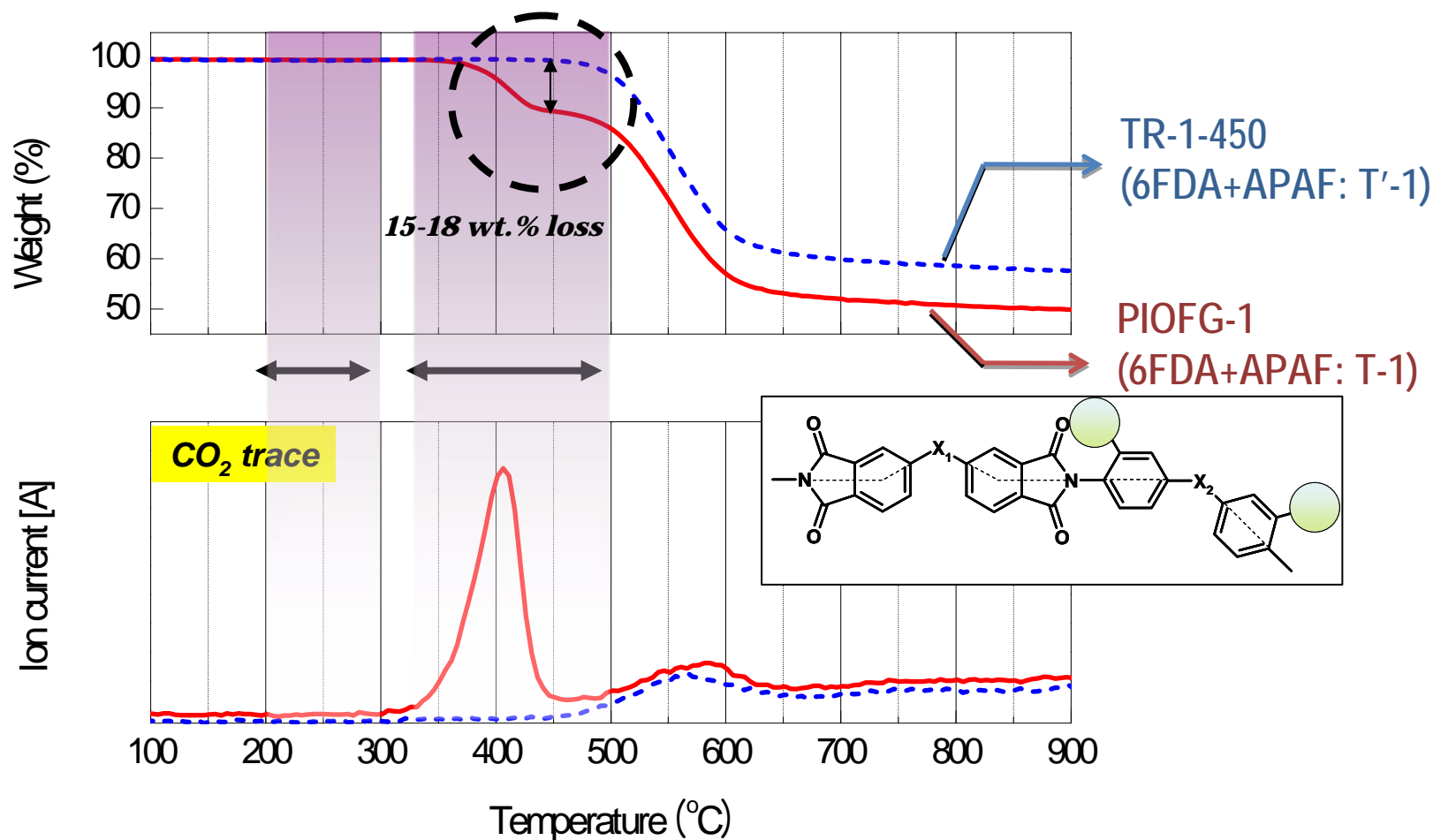


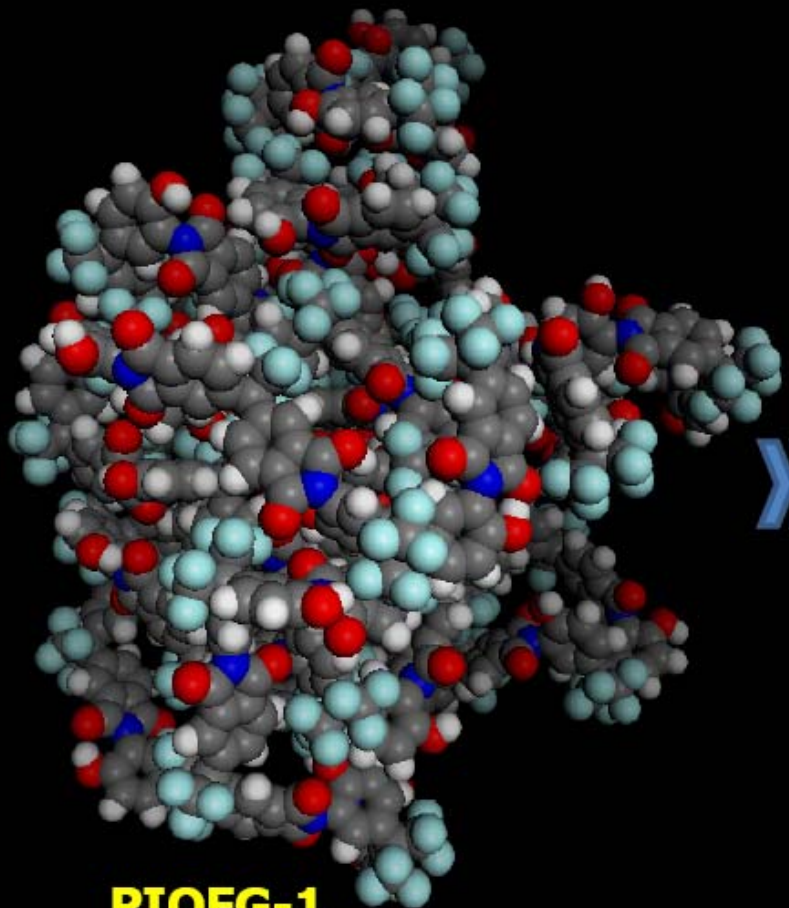
# Polyimides with Ortho - Positioned Functional Groups (PIOFG)



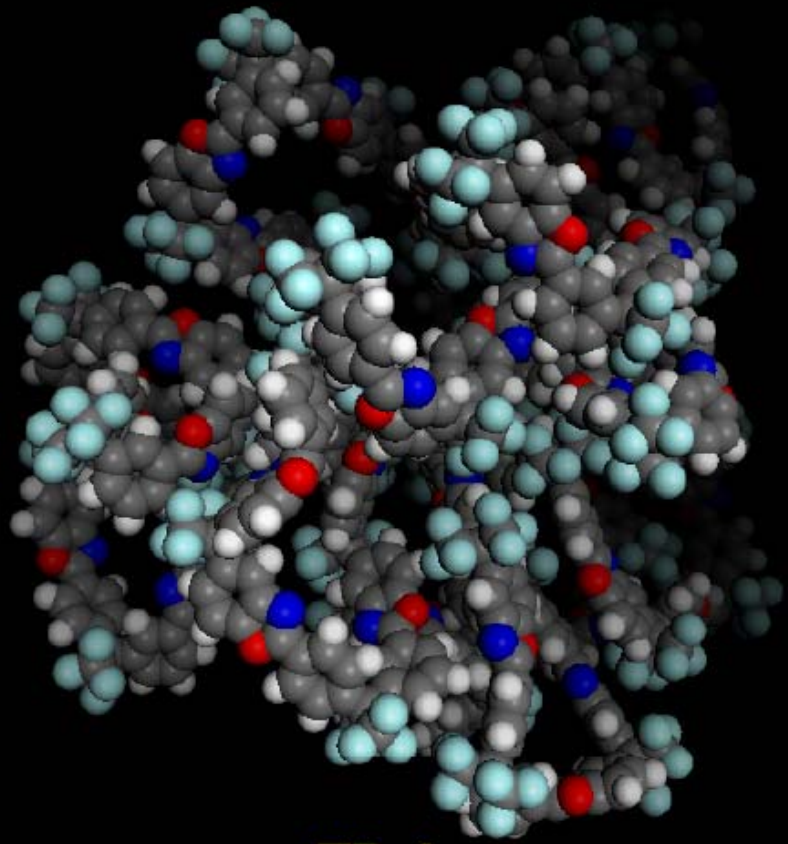
- Soluble
- High glass transition temperature (>350 °C)
- Thermally stable polymers for photoresist
- Non-linear optical (NLO) applications

# Thermal Conversion Process of PIOFG at High Temperature



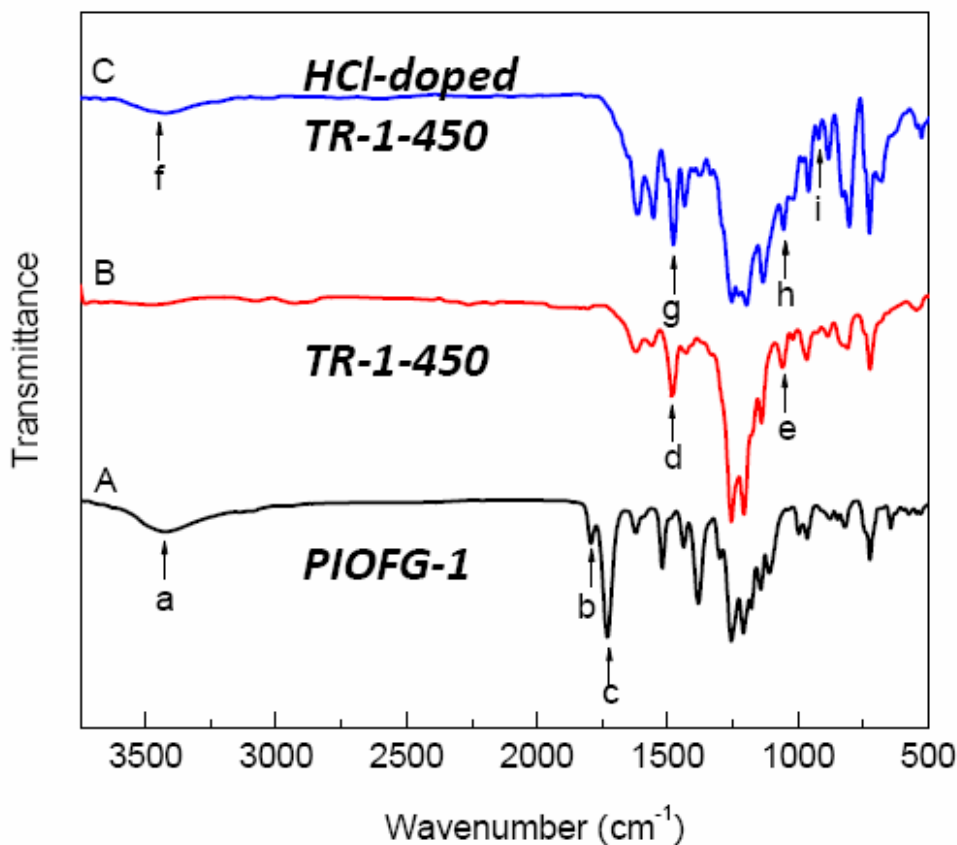


**PIOFG-1**

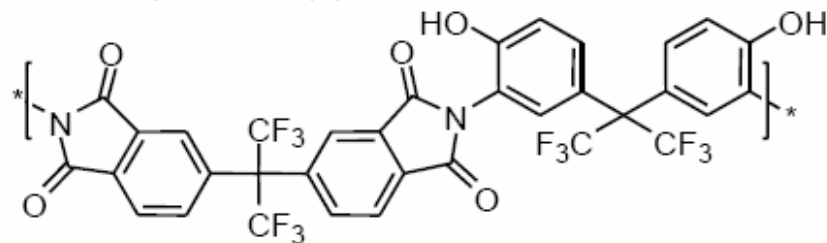


**TR-1**

# Spectroscopic Evidence for Thermal Conversion of PIOFG



FT-IR spectra of: (A) T-1, (B) T'-1, and (C) HCl-doped T'-1 samples (10 M HCl aqueous solution). (Key: *a.* = 3,400  $\text{cm}^{-1}$  (HO-phenylene), *b* = 1,788  $\text{cm}^{-1}$  (symmetric C=O stretching vibration), *c* = 1,720  $\text{cm}^{-1}$  (asymmetric C=O stretching vibration), *d*( = *g*) and *e*( = *h*) = 1,480 and 1,058  $\text{cm}^{-1}$ , respectively, (stretching vibration of benzoxazole ring), *f* = 3,450  $\text{cm}^{-1}$ (water), and *i* = 920  $\text{cm}^{-1}$  (Cl anion).)



**PIOFG-1**

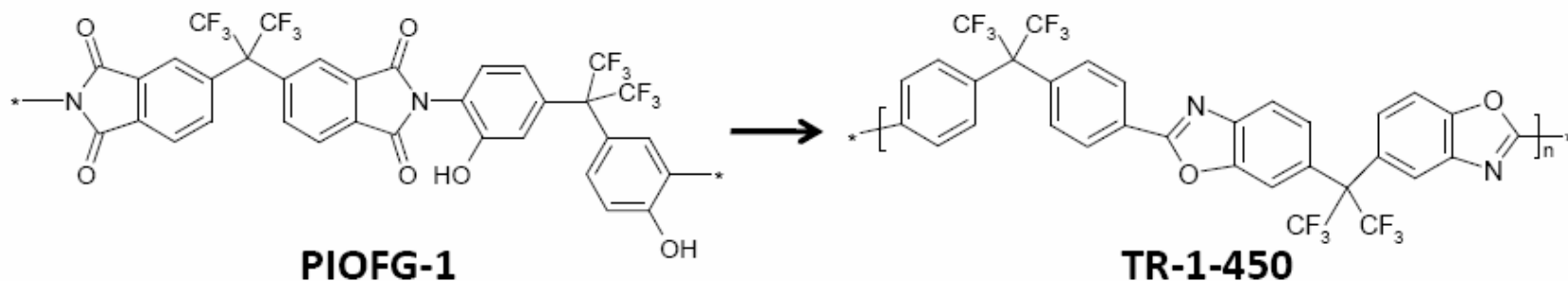
# Elemental Analysis

[Table S2] Elemental analysis data of PIOFG-1 and TR-1-450

Polymer	Formula	C (wt%)	H (wt%)	N (wt%)	O (wt%)	F (wt%)
PIOFG-1	$[\text{C}_{34}\text{H}_{14}\text{F}_{12}\text{N}_2\text{O}_6]_n$	$53.2 \pm 0.08$ (52.7)*	$1.87 \pm 0.06$ (1.82)*	$3.62 \pm 0.01$ (3.62)*	$11.3 \pm 0.22$ (11.3)*	$30.0 \pm 0.08$ (29.4)*
TR-1-350	-	$54.1 \pm 0.16$	$2.07 \pm 0.00$	$3.87 \pm 0.01$	$9.34 \pm 0.18$	$30.6 \pm 0.02$
TR-1-400	-	$55.2 \pm 0.01$	$2.02 \pm 0.01$	$4.05 \pm 0.00$	$7.23 \pm 0.03$	$31.5 \pm 0.04$
TR-1-450	$[\text{C}_{32}\text{H}_{14}\text{F}_{12}\text{N}_2\text{O}_2]_n$	$56.7 \pm 0.01$ (55.9)*	$1.93 \pm 0.02$ (2.06)*	$4.21 \pm 0.01$ (4.08)*	$4.89 \pm 0.12$ (4.66)*	$32.3 \pm 0.12$ (33.2)*

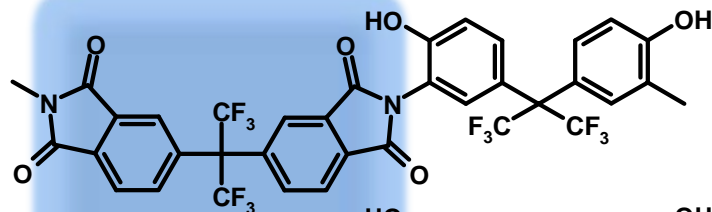
\*Calculated values

- Measured using ThermoFinnigan (Carlo Erba/Fison) EA1108
- Temperature: 1000 °C, (1060 °C for O)
- Catalyst:  $\text{WO}_3/\text{Cu}$ , (Nickel plated carbon, nickel wool, quartz turnings, soda lime, magnesium perchlorate anhydrous for O)
- Sample mass: 5 mg, (2 mg for O)
- Measured elements: C, H, N, O
- Standard material: BBOT (2,5-bis(5-tert-butyl-benzoxazole-2-yl) thiophene), (sulfanilamide for O)

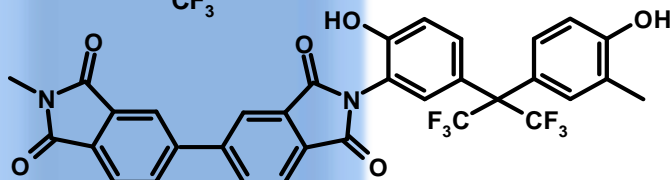


# Selected PIOFG Structures

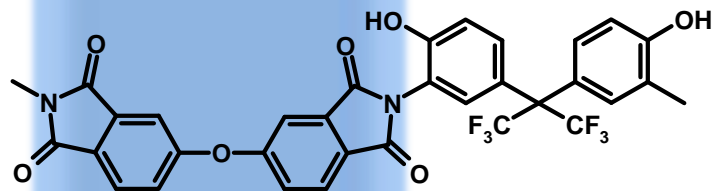
PIOFG-I



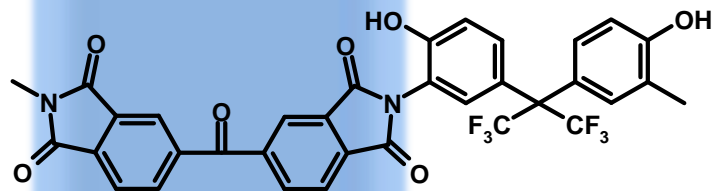
PIOFG-II



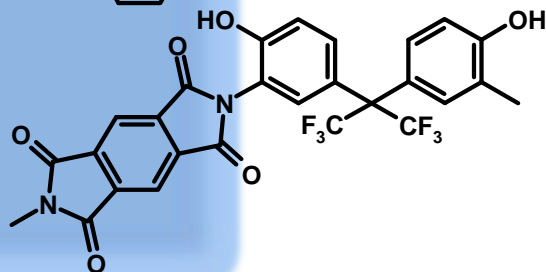
PIOFG-III



PIOFG-IV



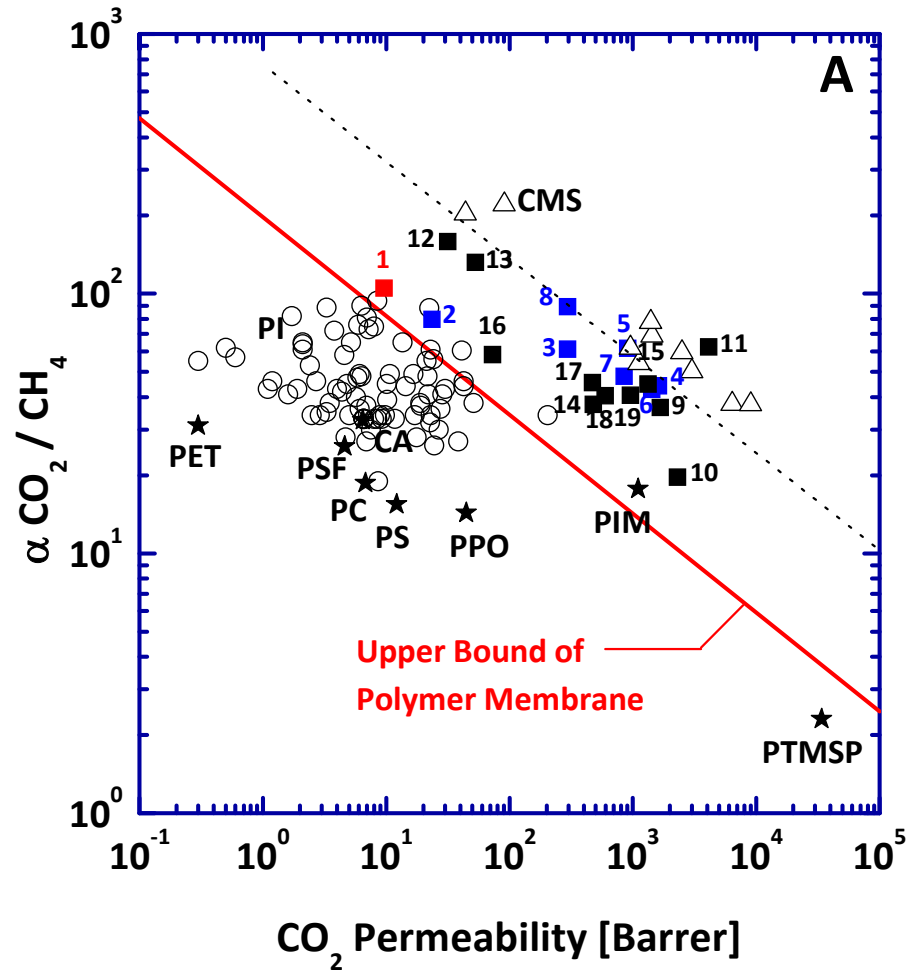
PIOFG-V



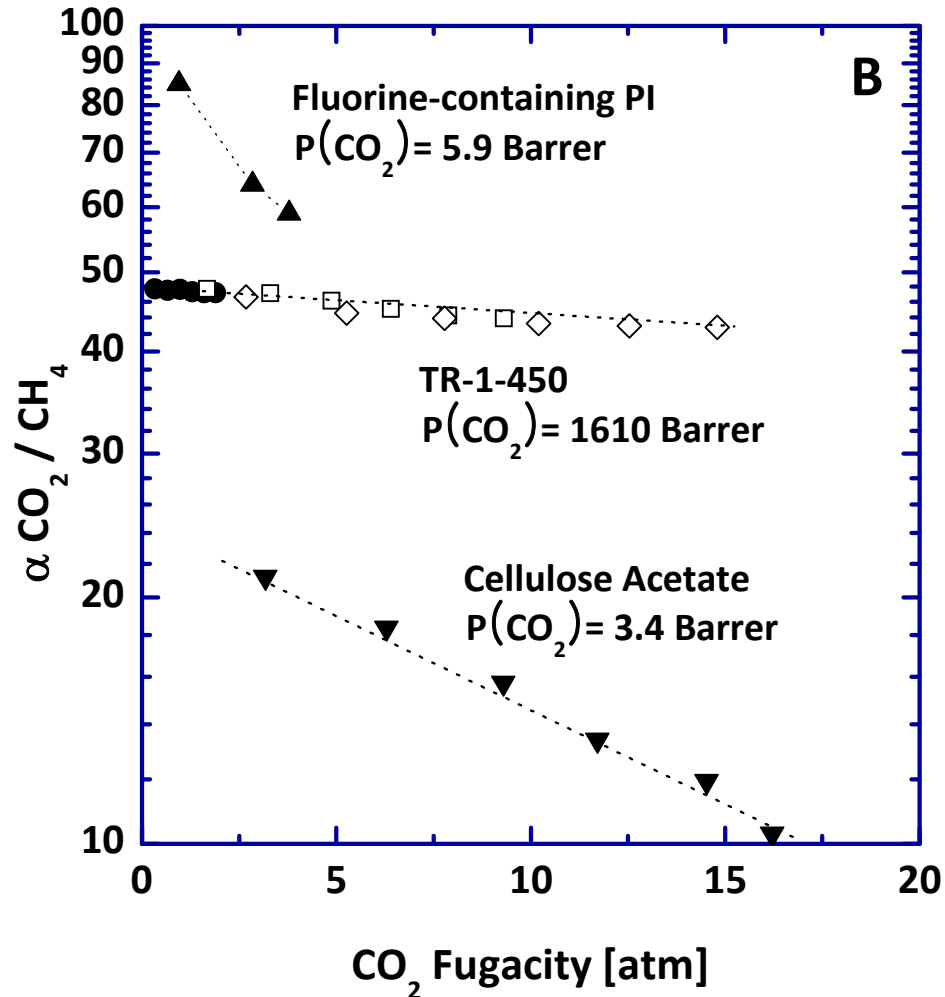
# Physical Properties of PIOFG and TR Polymers

Sample code	Density (g/cm <sup>3</sup> )	V (cm <sup>3</sup> /g)	V <sub>w</sub> (cm <sup>3</sup> /g)	FFV	Increment in FFV (%)	d-spacing (nm)
PIOFG-1	1.503	0.665	0.430	0.159	65	0.548
TR-1	1.293	0.773	0.439	0.263		0.600
PIOFG-2	1.453	0.688	0.459	0.134	64	0.546
TR-2	1.271	0.787	0.473	0.219		0.606
PIOFG-3	1.469	0.681	0.455	0.131	57	0.503
TR-3	1.304	0.767	0.469	0.205		0.611
PIOFG-4	1.482	0.675	0.457	0.120	102	0.539
TR-4	1.240	0.806	0.470	0.243		0.602
PIOFG-5	1.478	0.677	0.443	0.148	28	0.560
TR-5	1.362	0.734	0.457	0.190		0.698

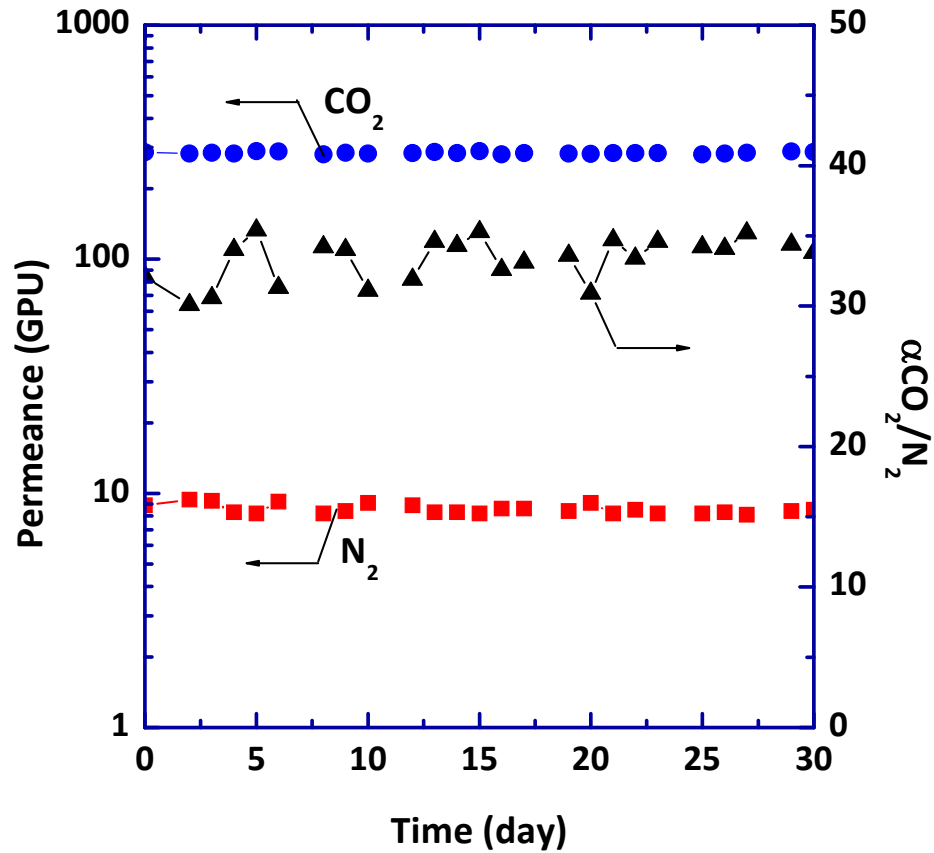
# CO<sub>2</sub>/CH<sub>4</sub> Separation



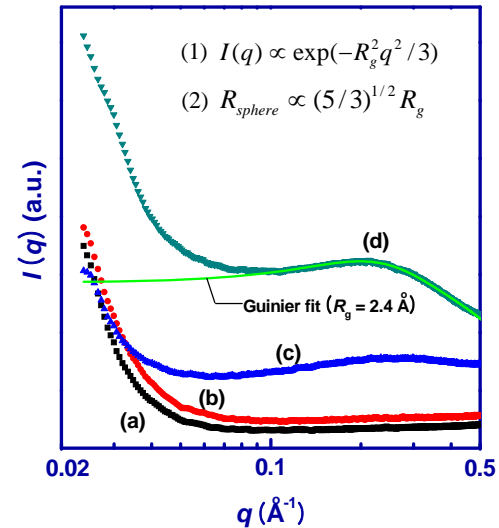
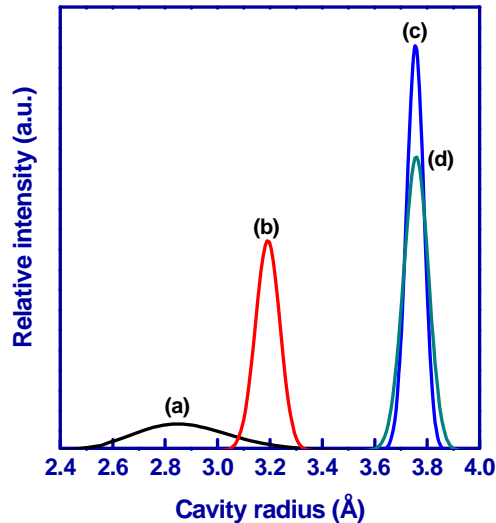
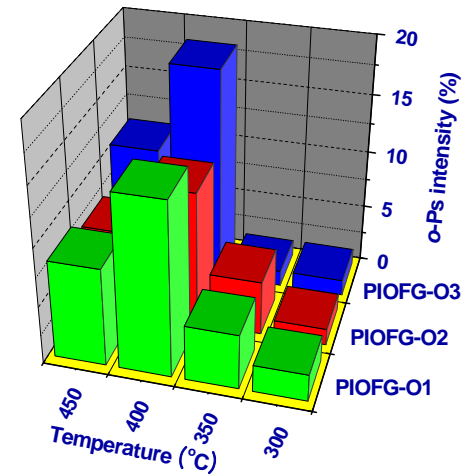
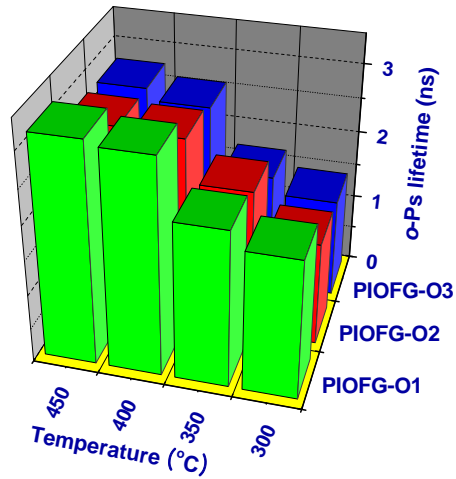
# Anti - Plasticization Behavior of TR Polymer Membranes



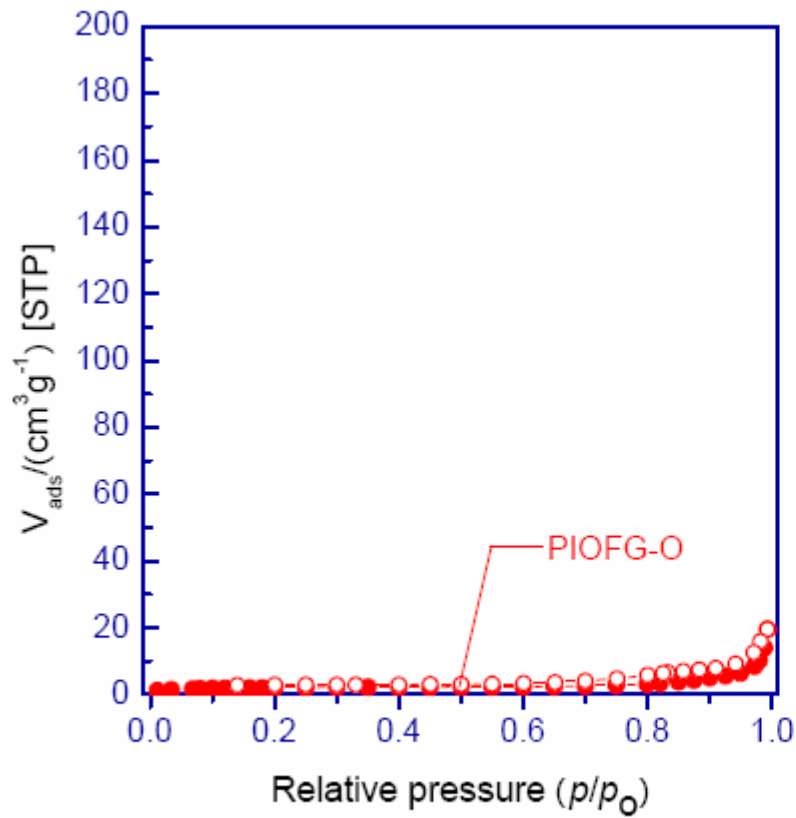
# CO<sub>2</sub>/N<sub>2</sub> Separation



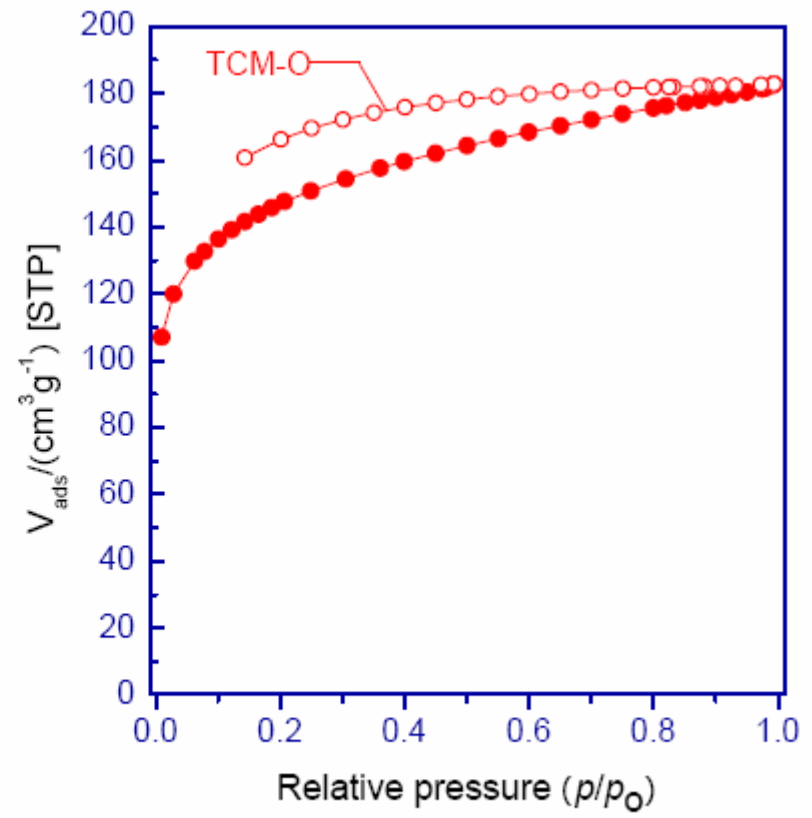
# Cavity Size Distribution of PIOFG and TR Polymers



# Microporous Character?

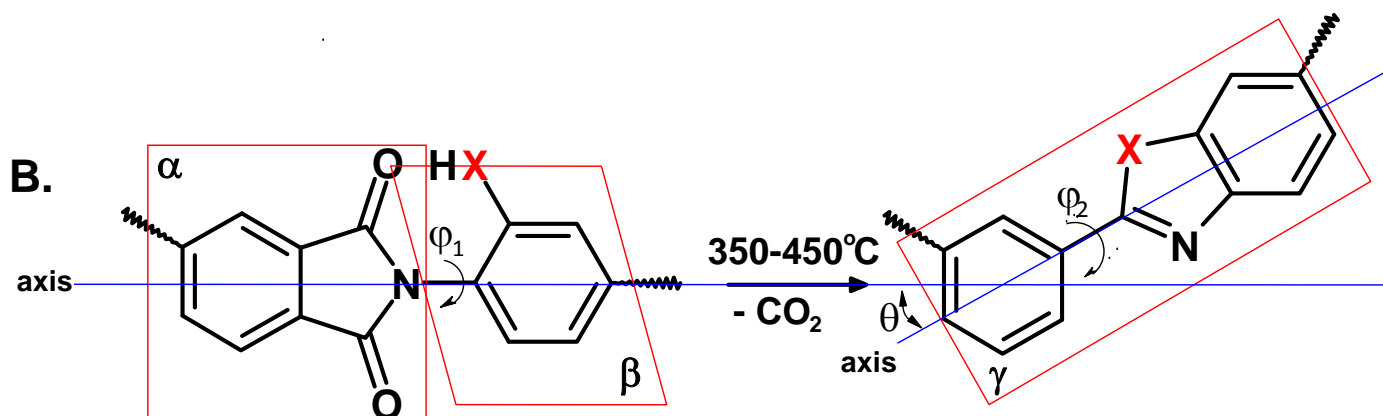
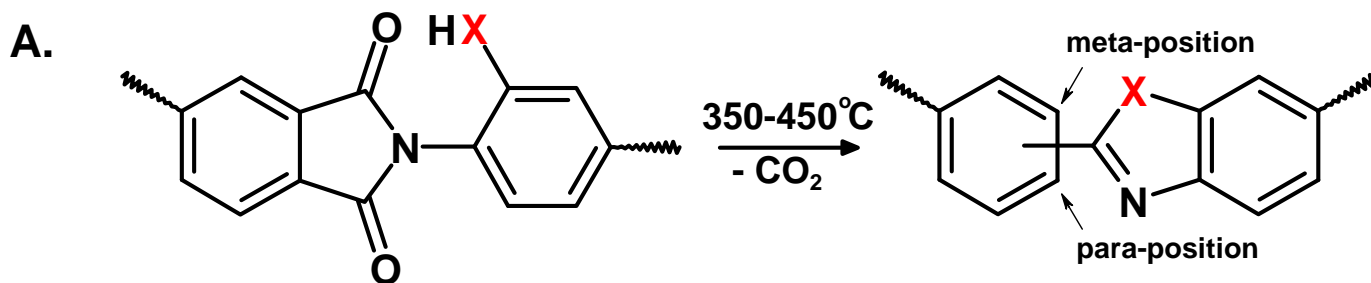


(A)

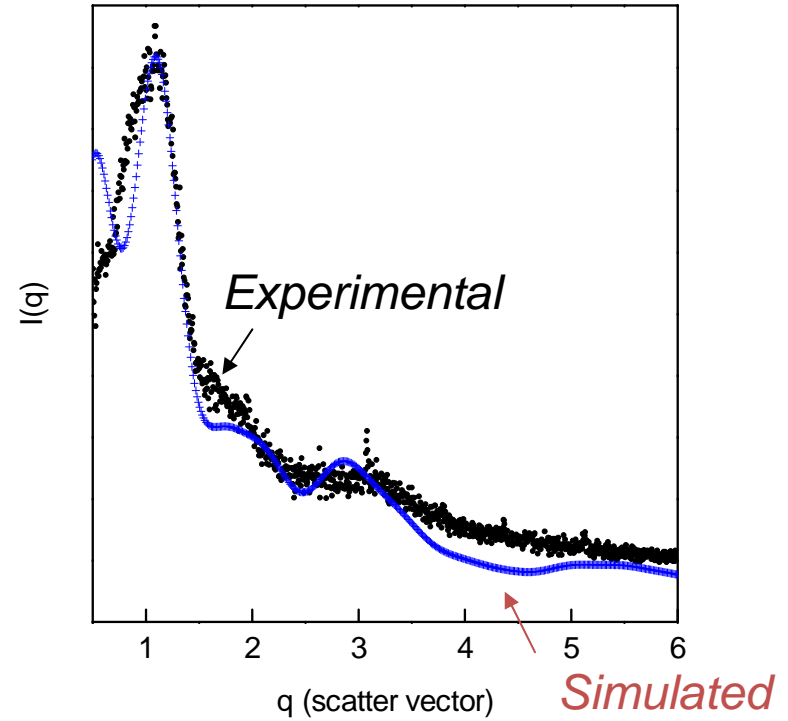
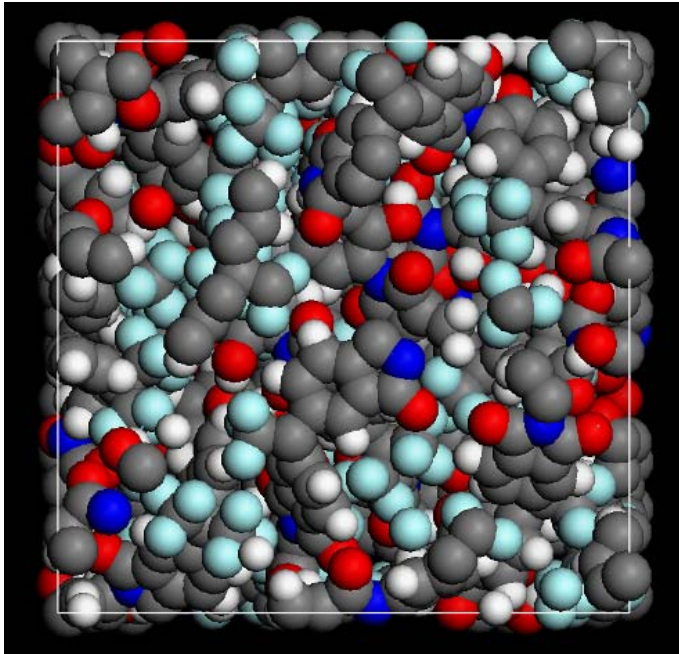


(B)

# Chain Rearrangement of Interst



# Simulated PIOFG - 1 Polymer Chain



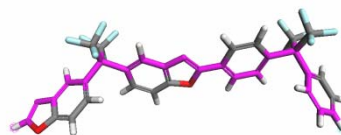
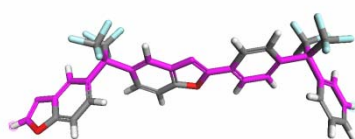
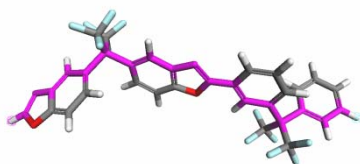
# Effect of Regioisomerism on Final Polymer Structures

*m-m* linked PBO

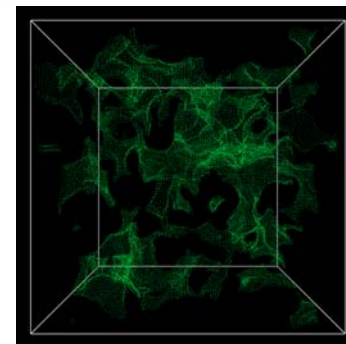
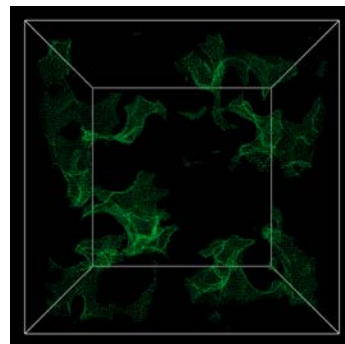
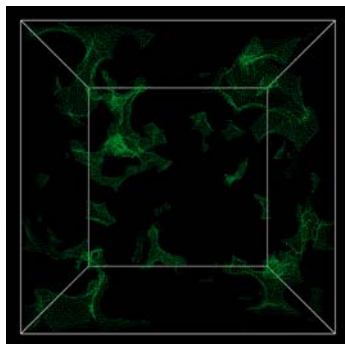
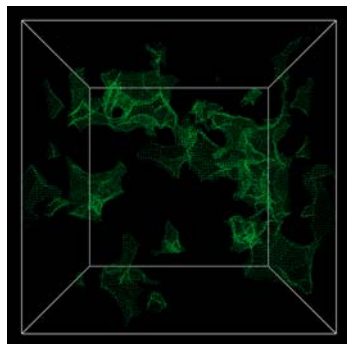
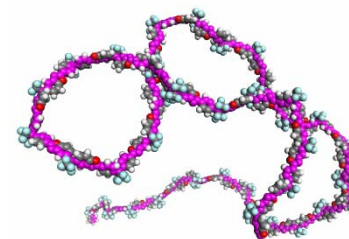
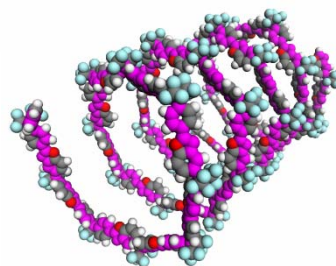
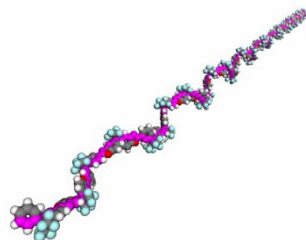
*m-p* (*p-m*) linked PBO

*p-p* linked PBO

*mixed* PBO

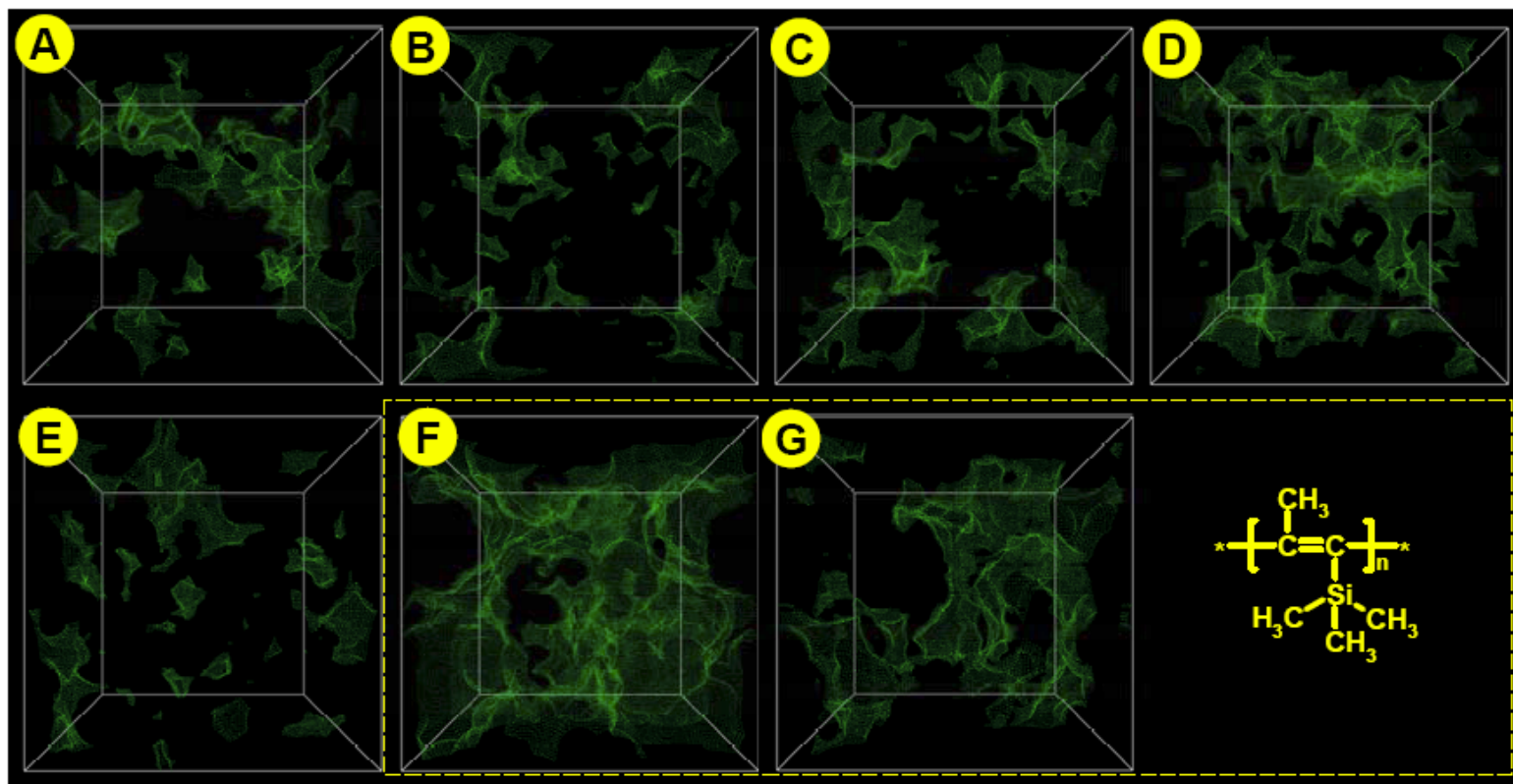


*m-m:p-p:m-p*  
*=1:1:2*



Length: 30.736 Å, Angle: 90 °

# Effect of Packing Density on Different Chain Conformations



A: m-m TCM-O,

B: m-p TCM-O,

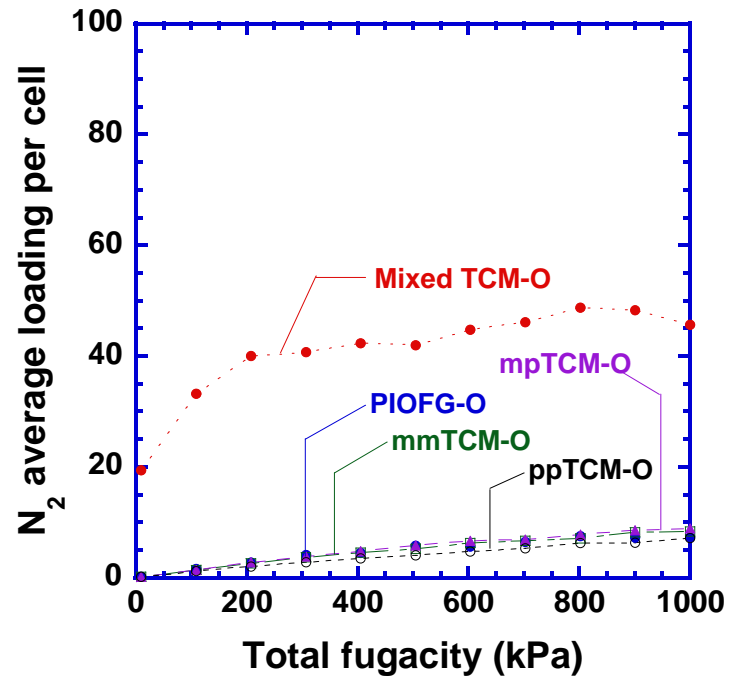
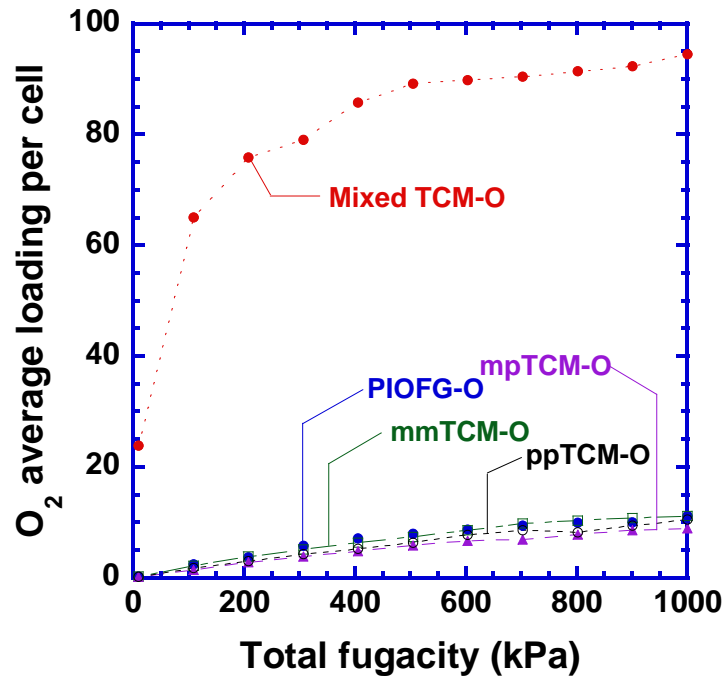
C: p-p TCM-O,

D: mixed TCM-O

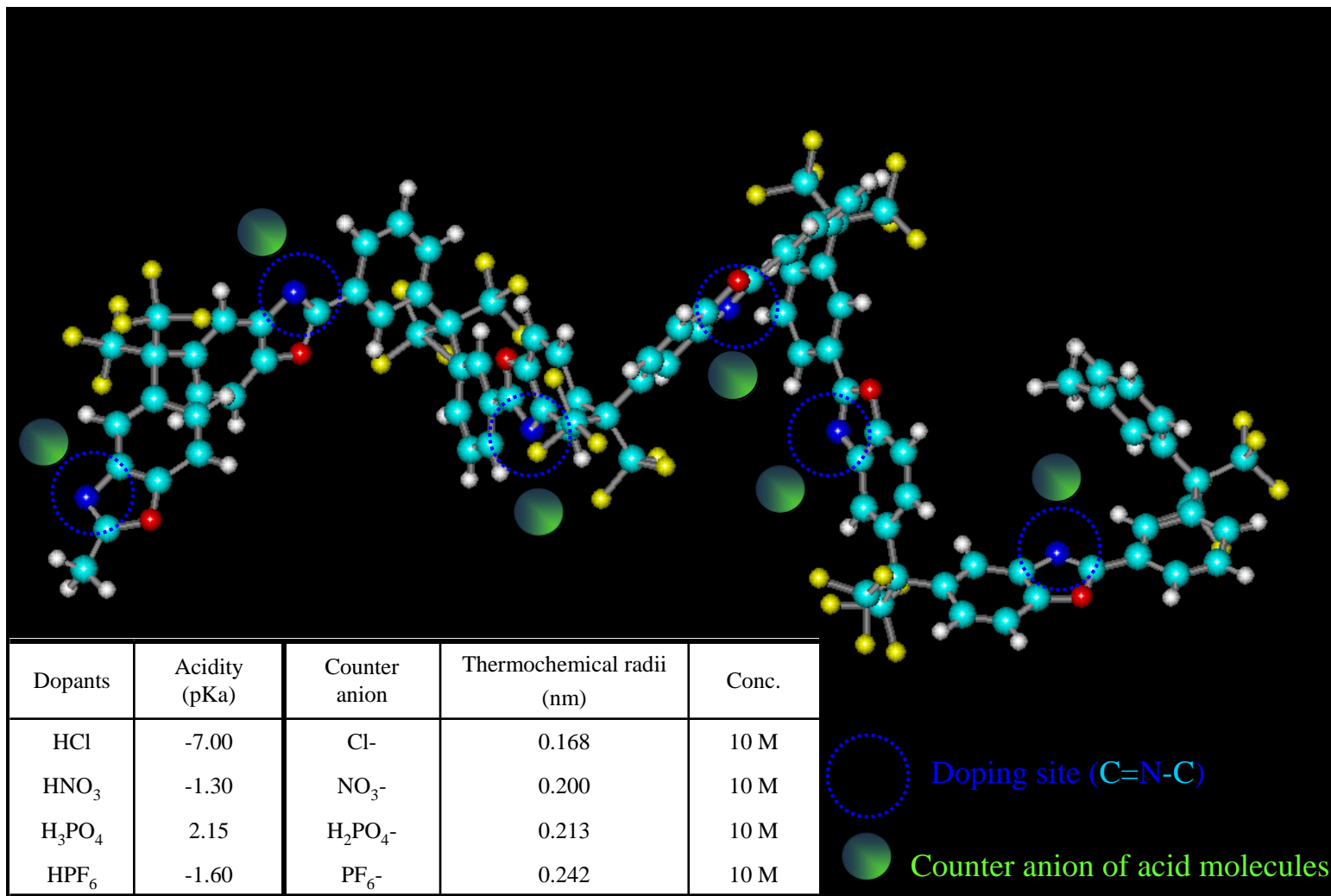
E: PIOFG-O (polyimide), F: pure PTMSP,

G: aged PTMSP

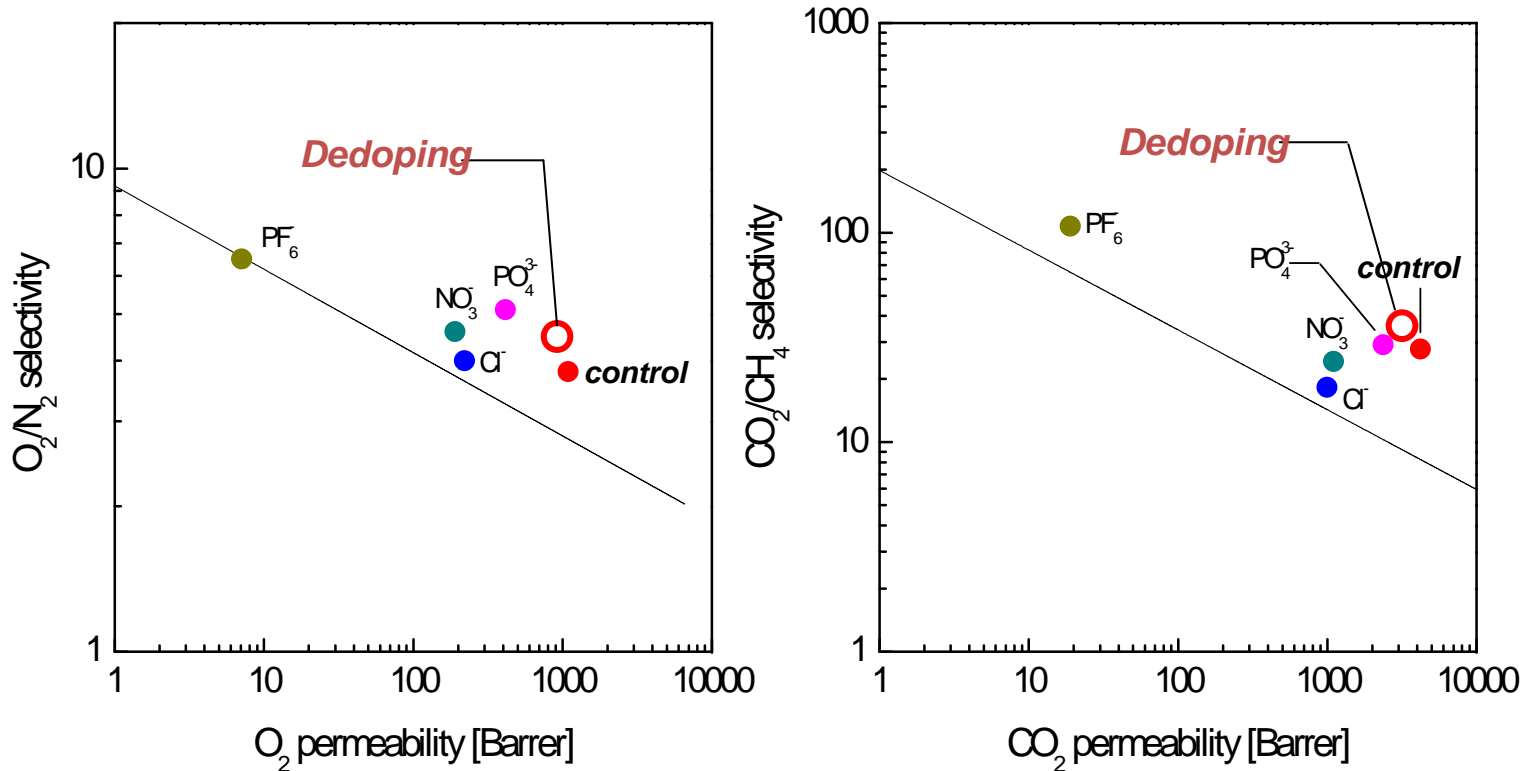
# Mixed Chain Conformation Increases the Amount of Sorption



# Control of Cavity Size By Dopants



# Change in Gas Permeability by Dopants



# Conclusions

- We showed a plastic, atypical free volume elements in dense, vitreous polymers can promote outstanding molecular and ionic transport along with enhanced separation performance.
- The unusual cavity (free volume elements) evolution can be systematically tailored using a “molecular rearrangement concept” which takes place in the glassy, amorphous, confined polymers.
- Free volume texture can be changed by controlling the degree of conversion, the flexibility of the original chain, and the insertion method of small dopant molecules.
- The current approach overcomes the molecular separation limit of conventional organic polymers and demonstrates the expanded applicability of this family of polymers in physical and chemical separations.

# Acknowledgements



**Carbon Dioxide Reduction  
& Sequestration R&D Center**



**Brain Korea**

**Collaboration between University of Texas Austin and CSIRO**

# Mechanical properties of PIOFG, TR polymers and CMS films at 25 °C

Sample	Tensile strength (MPa) <sup>g</sup>	Elongation percent at break (%) <sup>g</sup>
PIOFG-1	83	3.1
TR-1-350	87	3.8
TR-1-400	95	3.5
TR-1-450	98	3.9
CMS-1 <sup>a</sup>	42	0.4
CMS-2 <sup>b</sup>	52	0.3
CMS-3 <sup>c</sup>	34	0.6
PBO-1 <sup>d</sup>	98	10
PBO-2 <sup>d</sup>	99	11
PBO-3 <sup>d</sup>	52	4
PBO-4 <sup>d</sup>	77	8
PBO-5 <sup>e</sup>	110	4.5
PBO-6 <sup>f</sup>	89.6	10.9

<sup>a</sup> Carbon molecular sieve film derived from Kapton polyimide (carbonization temperature = 600 °C)

<sup>b</sup> Carbon molecular sieve film derived from Kapton polyimide (carbonization temperature = 800 °C)

<sup>c</sup> Carbon molecular sieve film derived from PIOFG-1 (carbonization temperature = 600 °C)

<sup>d</sup> Polybenzoxazole film (S1)

<sup>e</sup> Polybenzoxazole film (S2)

<sup>f</sup> Polybenzoxazole film (S3)

<sup>g</sup> Five specimens of each sample were tested. Standard deviation from the mean was within  $\pm 5\%$ .

# No Plasticization

